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# Effects of Metal–Organic Framework Membrane on Hydrogen Selectivity

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### Abstract

Hydrogen gas has attracted considerable attention as a promising candidate for future energy resources because of its eco-friendly characteristics; however, its highly combustible characteristics should be thoroughly examined to preclude potential disasters. In this regard, a highly sensitive method for the selective detection of  $H_2$  is extremely important. To achieve excellent  $H_2$  selectivity, the utilization of a metal–organic framework (MOF) membrane can physically screen interfering gas molecules by restricting the size of kinetic diameters that can penetrate its nanopores. This paper summarizes the various endeavors of researchers to utilize the MOF molecular sieving layer for the development of highly selective  $H_2$  sensors. Further, the review affords useful insights into the development of highly reliable  $H_2$  sensors.

Keywords : Gas sensor, Hydrogen sensor, Metal-organic frameworks, Nanoporous, Membrane

#### **1. INTRODUCTION**

Hydrogen gas (H<sub>2</sub>) has been regarded as one of the most attractive future energy resources because of its highly ecofriendly consumption mechanisms for energy generation [1]. Moreover, H<sub>2</sub> is well-known for its highly combustible characteristics, including low ignition energy, wide flammability range, and high combustion velocity [2]. Hence, to ensure the safe utilization of H<sub>2</sub>, the use of highly sensitive and selective H<sub>2</sub>-sensing technologies is advantageous. Related research investigations that have been conducted over the recent decades have found that the most effective strategies for H<sub>2</sub> detection include the utilization of Pd [3-6]. Palladium is well-known for forming palladium hydride (i.e., PdH<sub>x</sub>) after exposure to H<sub>2</sub>, resulting in resistance change (low concentration, <1%) or volume expansion (high concentration, >2%). These PdHx characteristics can be exploited for the development of highly selective H<sub>2</sub> sensors [7]. However, in actual applications, Pd-based H<sub>2</sub> sensors are exposed to ambient

air containing  $O_2$ , which can reduce the efficiency of  $H_2$ -sensing by blocking the active sites of Pd lattices [8]. Accordingly, various strategies to inhibit the degradation of Pd by  $O_2$  have been investigated. [9]

The utilization of a metal-organic framework (MOF) has been regarded as one of the most effective methodologies in this aspect; it can provide a highly selective membrane because of its easily controllable pore sizes. The MOF, which is composed of metal nodes and organic linkers, have been adopted in gas sensor applications i) as the main gas-sensing channel [10-13], ii) as a protective and selective membrane [14-18], or iii) as a pre-step material for MOF-derived gas-sensing materials [19-22]. However, the MOF itself is limited by its low conductivity and poor chemical stability with no distinct mechanism for selective H<sub>2</sub>-sensing [23]. Moreover, the MOF-derived materials are usually metal oxides that are advantageous for detecting volatile organic compounds but without a distinct mechanism for selective H<sub>2</sub>-sensing [24-27]. However, the MOF can provide a molecular sieving layer on top of gas-sensing materials with precisely controllable pore sizes. The MOF can be utilized for the physical screening of target gas molecules according to their kinetic diameter; this can be beneficial for achieving H<sub>2</sub> selectivity.

This paper summarizes the endeavors of researchers to utilize MOF as a molecular sieving layer for the development of a highly selective Pd-based  $H_2$  sensor. These endeavors demonstrate that the use of MOF-based protective and molecular sieving layers can be a highly effective strategy for enhancing the  $H_2$ -sensing performance and lifetime of sensors. This review is anticipated to

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Fig. 1. Schematic of metal-organic framework

provide effective and useful insight into the development of highly reliable and potential  $H_2$  sensors.

# 2. UTILIZATION OF METAL–ORGANIC FRAMEWORK MEMBRANE

## 2.1 Pore Size Effects of MOF Membrane

The MOF is comprised of metal nodes and organic linkers, as

shown in Fig. 1, and has extremely large surface areas, highly porous structures, and various morphologies [28]. In particular, the appropriate choice of metal species and organic linkers can result in the formation of nanopores with preferred diameters that may be used for various MOF separation membranes [29]. In this regard, the MOF membrane with a pore size larger than the kinetic diameter of  $H_2$  but smaller than those of other interfering gas molecules can lead to a highly selective detection of  $H_2$ .

M. Drobek *et al.* fabricated gas sensors based on pristine ZnO nanowires (NW) and ZnO NWs encapsulated by a thin zeolite



Fig. 2. (a) Schematic of the synthesis steps yielding the ZIF-membrane encapsulated ZnO nanowires (ZnO@ZIF-8 NWs). (b) Plane-view FESEM image of networked ZnO nanowires, (c) ZnO@ZIF-8 composite nanowires, and (d) TEM images of ZnO@ZIF-8 composite nanowires and the corresponding STEM/EDX cartography for Zn, N, and C elements. (e-g) Responses of the pristine ZnO nanowires and ZnO@ZIF-8 composite nanowires in contact with 10, 30, and 50 ppm single gas concentration: (e) H₂, (f) C<sub>7</sub>H<sub>8</sub>, and (g) C<sub>6</sub>H<sub>6</sub>. Reprinted with permission from [14]. Copyright © 2016 American Chemical Society.

imidazolate framework (ZIF)-8 molecular sieve membrane [14]; Fig. 2(a) shows the preparation method for the ZIF membraneencapsulated ZnO NWs. Using the solvothermal method, the concentration of organic linkers in the reaction solution was precisely controlled for the optimized conversion of ZnO into ZIF-8. Fig. 2(b) and (c) show the scanning electron microscopy (SEM) images of the resultant ZIF membrane-encapsulated ZnO NWs. These images show the more closely stacked ZnO NWs after the solvothermal treatment for ZIF-8 encapsulation. The transmission electron microscopy (TEM) images (Fig. 2(d)) clearly show the ZIF layer encapsulating the ZnO NWs. After the optimization of fabrication procedures, the pristine ZnO NWs and ZIF-encapsulated ZnO NWs were exposed to three different target gases, including  $H_2$ ,  $C_7H_8$ , and  $C_6H_6$ . As shown in Fig. 2(e)–(g), the gas responses to all three target gases decreased after the ZIF encapsulation, which certainly reduced the accessibility of target gas molecules to the ZnO NWs. The response to the 50 ppm  $H_2$  remained at 1.44, whereas the responses to the 50 ppm  $C_7H_8$  and  $C_6H_8$  were practically reduced to negligibility. The prepared ZIF-8 encapsulation layer has a small portal cavity size (3.4 Å), and the kinetic diameters of  $C_7H_8$  and  $C_6H_6$  are 5.92 and 5.27 Å, respectively. Therefore, both  $C_7H_8$  and  $C_6H_6$  could not penetrate the ZIF-8 encapsulation layer; accordingly, they could not affect the surface of ZnO NWs. Meanwhile,  $H_2$ , which has a kinetic diameter of



Fig. 3. (a-c) SEM images of ZnO NRAs, ZnO@ZIF-8 NRAs, ZnO@ZIF-71 NRAs: (a) for vertically-aligned ZnO NRAs, (b) for ZnO NRAs coated with ZIF-8. (c) for ZnO NRAs coated with ZIF-71. (d) Gas concentration gradient response curves of the ZnO NRAs, ZnO@ZIF-8 NRAs and ZnO@ZIF-71NRAs: these three kinds of gas sensors are exposed in hydrogen, (10, 50, 100, 150, 200 ppm) at working temperature of 250°C. (e) Gas sensing response of ZnO NRAs, ZnO@ZIF-8 NRAs and ZnO@ZIF-71 NRAs sensors to 50 ppm of different gases at 250°C. (f) The mechanism of using the difference between the pore sizes of ZIF-8 and ZIF-71 and the gas molecular sizes to select gases passing the ZIFs membrane of ZnO@ZIF NRAs. Reprinted with permission from [15]. Copyright © 2018 Elsevier B.V.

2.89 Å, could easily penetrate the ZIF-8 encapsulation layer and reach the ZnO NWs, resulting in the highly selective detection of  $H_2$ . This work demonstrates that the precise control of the MOF pore size can lead to the achievement of the desired gas selectivity depending on the kinetic diameter of the target gas molecules.

T. Zhou et al. investigated the effect of MOF pore sizes on gas selectivity by preparing two different MOF membranes (ZIF-8 and ZIF-71) on a ZnO nanorod array [15]. The SEM images of the hydrothermally prepared ZnO nanorods, ZnO encapsulated with ZIF-8, and ZnO encapsulated with ZIF-71 are shown in Fig. 3(a)-(c), respectively. Compared with pristine ZnO nanorods, the densely coated MOF membranes are clearly identifiable. The prepared samples were then exposed to five different target gases: NH<sub>3</sub>, H<sub>2</sub>, C<sub>2</sub>H<sub>5</sub>OH, CH<sub>3</sub>COCH<sub>3</sub>, and C<sub>6</sub>H<sub>6</sub>. Fig. 3(d) shows the gas response to H<sub>2</sub> as a function of H<sub>2</sub> concentration, and Fig. 3(e) presents the gas responses of the prepared gas sensors to all target gases. Compared with the pristine ZnO nanorods, the ZIF-8encapsulated ZnO nanorods exhibited practically the same gas response toward NH<sub>3</sub> and H<sub>2</sub>; the gas response toward C<sub>2</sub>H<sub>5</sub>OH, CH<sub>3</sub>COCH<sub>3</sub>, and C<sub>6</sub>H<sub>6</sub> significantly decreased. The ZIF-71encapsulated ZnO nanorods only exhibited a significant decrease in gas response toward C<sub>6</sub>H<sub>6</sub> and maintained its gas responses to the other four target gases. The pore sizes of ZIF-8 and ZIF-71 are 3.4 and 4.8 Å, respectively; hence, the foregoing results indicate that the pore size of the ZIF membrane can determine the gas selectivity of ZnO nanorods. The NH<sub>3</sub>, H<sub>2</sub>, C<sub>2</sub>H<sub>5</sub>OH, CH<sub>3</sub>COCH<sub>3</sub>, and C<sub>6</sub>H<sub>6</sub> gases have kinetic diameters of 2.90, 2.89, 4.53, 4.60, and 5.85 Å, respectively. Therefore, target gases with kinetic diameters smaller than that of ZIF-8 (NH<sub>3</sub> and H<sub>2</sub>) could penetrate the ZIF-8 membrane, affecting the ZnO nanorods. Those with larger kinetic diameters (C<sub>2</sub>H<sub>5</sub>OH, CH<sub>3</sub>COCH<sub>3</sub>, and C<sub>6</sub>H<sub>6</sub>) could penetrate the ZIF-8 membrane, yielding low gas responses. For ZIF-71, only  $C_6H_6$  exhibits a kinetic diameter larger than the pore size of the ZIF-71 membrane; thus, it is the only target gas that exhibits a decrease in gas response (Fig. 3(f)). Therefore, the precise control of the pore sizes of the MOF membrane is a potentially effective strategy for the development of gas sensors with the preferred selectivity.

2.2 Effects of MOF Membrane on Pd-based  $H_2$  Sensors

As the most widely adopted MOF membrane for the selective detection of  $H_2$  owing to its suitable cavity size (3.4 Å), researchers have endeavored to utilize ZIF-8 for Pd-based gas sensors.

M. Weber et al. developed highly sensitive and selective H<sub>2</sub>

sensors based on ZnO NWs decorated with Pd nanoparticles; they also fabricated an MOF nanomembrane molecular sieve (ZIF-8) [16]. Fig. 4(a) shows the schematic of the preparation of vaporgrown ZnO NWs on interdigitated electrodes (IDEs), the successive decoration of Pd nanoparticles, and the conversion of the ZnO surface into ZIF-8. The prepared gas sensors based on pristine ZnO NWs, Pd-decorated ZnO NWs, and ZIF-8encapsulated Pd-decorated ZnO NWs were then exposed to five different target gases: H<sub>2</sub>, C<sub>6</sub>H<sub>6</sub>, C<sub>2</sub>H<sub>5</sub>OH, C<sub>7</sub>H<sub>8</sub>, and CH<sub>3</sub>COCH<sub>3</sub>. Fig. 4(b) shows the gas responses of the prepared gas sensors toward H<sub>2</sub> as a function of H<sub>2</sub> concentration. The Pd decoration significantly enhanced the gas responses toward H<sub>2</sub> compared with pristine ZnO NWs. These responses decreased to a certain level (2.63 for 50 ppm H<sub>2</sub>) after the ZIF-8 encapsulation was applied because of the decreased gas accessibility. Fig. 4(c) shows the gas responses of the prepared gas sensors toward the other four target gases depending on the gas concentration. With the Pd decoration, the gas responses toward these four target gases ( $C_6H_6$ , C<sub>2</sub>H<sub>5</sub>OH, C<sub>7</sub>H<sub>8</sub>, and CH<sub>3</sub>COCH<sub>3</sub>) also slightly increased due to chemical sensitization effects. After the ZIF-8 encapsulation, however, all gas responses toward these four target gases were remarkably suppressed to practically negligible levels, resulting in the highly selective detection of  $H_2$ . Fig. 4(d) shows the exact mechanisms of the gas-sensing reaction and the effect of ZIF-8 encapsulation. Upon H<sub>2</sub> exposure, the Pd-decorated ZnO captured H<sub>2</sub> through the formation of PdH<sub>x</sub> for the enhanced gas responses. By contrast, the other target gases only relied on the surface reaction through charge exchange with the surface-adsorbed oxygen species [30]. The ZIF-8 encapsulation provided a porous membrane for the exclusive penetration of H<sub>2</sub> for the highly selective detection of H<sub>2</sub> through the PdH<sub>x</sub> formation. Although this work demonstrated the excellent feasibility of the ZIF-8 membrane for the highly selective detection of H<sub>2</sub> by Pd-based H<sub>2</sub> sensors, the prevention of Pd degradation by O2 could not be confirmed. The main conduction channel was ZnO, and the lift time of the H<sub>2</sub> sensors was dependent on the ZnO stability, whereas the role of Pd was limited to the catalytic effects.

W.-T. Koo *et al.* prepared  $H_2$  sensors based only on Pd NWs, which are suitable for investigating the effect of MOF encapsulation on Pd stability [17]. Fig. 5(a)–(d) show the SEM images of pristine Pd NWs and ZIF-8 encapsulated Pd NWs (4 h treatment). The Pd NWs were prepared by photolithography, and ZIF-8 was synthesized by solution methods. The authors manipulated the deposition time of ZIF-8 on Pd NWs (2, 4, and 6 h) to control the ZIF-8 encapsulation layer thickness. The SEM images show that after 4 h of the ZIF-8 deposition, the Pd NWs were completely



● H<sub>2</sub> ● H<sub>2</sub>O ● Adsorbed oxygen molecules Re: C<sub>6</sub>H<sub>6</sub>, C<sub>2</sub>H<sub>5</sub>OH, C<sub>7</sub>H<sub>6</sub>, CH<sub>3</sub>COCH<sub>3</sub> gas Re-O: Interacted gas

Fig. 4. (a) Schematic representation of the three key steps enabling the synthesis of the novel gas-sensing device. (b) Calibration curves of bare ZnO, Pd/ZnO, and ZIF-8/Pd/ZnO NW gas sensors to 10, 30, and 50 ppm H<sub>2</sub> gas. (c) Calibration curves of bare ZnO, Pd/ZnO, and ZIF-8/Pd/ZnO NW gas sensors to 10, 30, and 50 ppm H<sub>2</sub>, C<sub>6</sub>H<sub>6</sub>, C<sub>2</sub>H<sub>3</sub>OH, C<sub>7</sub>H<sub>8</sub>, and CH<sub>3</sub>COCH<sub>3</sub> interfering gases. (d) Schematic representation of the sensing mechanisms for a) pristine ZnO NW, b) Pd/ZnO NW and c) ZIF-8-coated Pd/ZnO NW sensors. Reprinted with permission from [16]. Copyright © 2018 American Chemical Society.

encapsulated by ZIF-8. The prepared  $H_2$  sensors based on Pd NWs were then exposed to different concentrations of  $H_2$  (Fig. 5(e) and (f)). The pristine Pd NWs exhibited  $H_2$ -sensing characteristics at high concentrations; however, they did not exhibit a reliable operation toward  $H_2$  at low concentrations.

Compared with pristine Pd NWs, the ZIF-8-encapsulated Pd NWs still exhibited  $H_2$ -sensing characteristics at low concentrations. The authors assumed that the unreliable  $H_2$ -sensing characteristics by the pristine Pd NWs were caused by the  $O_2$  adsorption on the surface of Pd NWs. The reduction in the active sites on the Pd

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Fig. 5. SEM of Pd NWs (a, b), and Pd NWs@ZIF-8\_4 h (c, d), on the glass substrate. Low-magnification SEM images of the samples with patterned Pd NWs (a gap of 10 μm) (a, c) and high-magnification SEM images of the samples (b, d). (e, f) Dynamic baseline resistance transitions in the concentration range of 0.02–1% [H<sub>2</sub>] at RT. (e) Pd NWs, and (f) Pd NWs@ZIF-8\_4 h. (g-i) Schematic illustration of accelerated hydrogen sensing properties of Pd NWs@ZIF-8. (g) Sensing model for Pd NWs without ZIF-8 membrane and (h) sensing model for Pd NWs@ZIF-8\_6. (i) Schematic illustrate of the Pd NW, Pd NWs@ZIF-8\_2 h, Pd NWs@ZIF-8\_4 h, and Pd NWs@ZIF-8\_6 h. Reprinted with permission from [17]. Copyright © 2017 American Chemical Society.

NWs by the O<sub>2</sub> adsorption limited the H<sub>2</sub>-sensing capability of Pd NWs at low concentrations. With ZIF-8 encapsulation, however, the O<sub>2</sub> adsorption can be physically prevented because the kinetic diameter of  $O_2$  (3.45 Å) is larger than the pore size of ZIF-8 (3.4 Å), resulting in the reliable  $H_2$  sensing even at a low concentration. In addition, it was found that the response and recovery rates increased significantly after the ZIF-8 encapsulation. Different from other previously reviewed H<sub>2</sub> sensors, the main sensing materials are Pd NWs and not semiconducting materials, such as ZnO. Therefore,  $H_2$  detection is mainly determined by the PdH<sub>x</sub> formation and not the oxygen absorbate-mediated charge transfer, which is a well-known gas-sensing mechanism of metal oxide semiconductors [31]. As a result, the prevention of  $O_2$ adsorption could lead to a highly efficient PdH<sub>x</sub> formation and significantly improved reaction and recovery rates. Moreover, the authors attributed the acceleration effect of ZIF-8 to the adsorption and desorption rates of H2 on Pd as a result of surface enhancement and bulk reactivity. These interpretations are well illustrated in Fig. 5(g)–(i). This work clearly demonstrated that the molecular sieving MOF membrane not only ensured the  $H_2$  selectivity but also contributed to reliable operation and stability over a wider range of  $H_2$  concentrations, contributing to a longer sensor lifetime.

#### **3. CONCLUSIONS**

In summary, the utilization of a nanoporous MOF membrane as a molecular sieving layer on  $H_2$ -sensing materials including metal oxide semiconductors and Pd-based materials can lead to the realization of highly selective detection of  $H_2$ . The potential  $H_2$ sensing characteristics may be attributed to the pore-size dependent molecular sieve membrane that can selectively allow the penetration of  $H_2$  molecules with small kinetic diameters. Moreover, the MOF membrane can prevent the  $O_2$  adsorption on active sites on the Pd surface for reliable operation and longer lifetime. Lastly, the effect of the MOF membrane on the acceleration of adsorption and desorption of  $H_2$  molecules on the Pd surface are expected.

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