

# High Sensitivity and Selectivity of Array Gas Sensor through Glancing Angle Deposition Method

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## Abstract

In this study, we propose an array-type gas sensor with high selectivity and response using multiple oxide semiconductors. The sensor array was composed of SnO<sub>2</sub> and In<sub>2</sub>O<sub>3</sub>, and the detection characteristics were improved by using Pt, Au, and Pd catalysts. All samples were deposited directly on the Pt interdigitated electrode (IDE) through the e-beam evaporator glancing angle deposition (GAD) method. They grew in the form of well-aligned nanorods at off-axis angles. The prepared SnO<sub>2</sub> and In<sub>2</sub>O<sub>3</sub> nanorod samples were exposed to CH<sub>3</sub>COCH<sub>3</sub>, C<sub>7</sub>H<sub>8</sub>, and NO<sub>2</sub> gases in a 300°C dry condition. Au-decorated SnO<sub>2</sub>, Au-decorated In<sub>2</sub>O<sub>3</sub>, and Pd-decorated In<sub>2</sub>O<sub>3</sub> exhibited high selectivity for CH<sub>3</sub>COCH<sub>3</sub>, C<sub>7</sub>H<sub>8</sub>, and NO<sub>2</sub>, respectively. They demonstrated a high detection limit of the sub ppb level computationally. In addition, measurements from each sensor were executed in the 40% relative humidity condition. Although there was a slight reduction in detection response, high selectivity and distinguishable detection characteristics were confirmed

**Keywords** : Gas sensors, Metal oxide semiconductors, Nanorod, SnO<sub>2</sub>, In<sub>2</sub>O<sub>3</sub>

## 1. INTRODUCTION

Exhaled human breath contains several different compounds, including a variety of volatile organic compounds (VOCs) [1,2]. The number of VOCs varies depending on the state of the human body. Expiration analysis is an important research field that has recently received tremendous interest due to the advancement of analysis technology and nanotechnology. Analysis of exhaled breath is a method that has been used for disease diagnosis since ancient times and was popular in the era of Hippocrates [3]. In general, highly sensitive gas sensors are a good option for the early diagnosis of diseases, such as lung cancer [4], diabetes [5], congestive heart failure [6], and asthma [7]. VOCs are mainly in the range of ppm–ppt (part per million to part per trillion); hence, laboratory methods such as GC–MS (gas chromatography–mass spectroscopy) [8] and PTR(proton-transfer-reaction) are used for

exhaled human breath detection. However, the following methods are limited in their efficacy for disease diagnosis owing to disadvantages, such as high cost, bulkiness, high power consumption, and low performance.

However, a resistive sensor using a metal oxide semiconductor (MOS) displays advantages, such as low power consumption, low cost, scalable manufacturing, and excellent stability. In particular, a fast detection speed enables real-time monitoring [9-11]. In fact, numerous exhalation diagnostic sensors have been developed by applying various nanostructures, such as nanorods, thin films, nanotubes, and nanowires [12,13]. The nanostructure-based sensor exhibits excellent performance, such as high surface/volume ratio, low defects, and simple integration of chip devices. However, these methods possess several limitations, such as multiple-step process and poor contact. In addition, exhaled human breath contains many kinds of VOCs; the problem of selectivity also needs to be solved. In the case where a single sample is used, limitations exist, such as classification of a specific gas and high selectivity.

In this study, we propose an array-type gas sensor using an oxide semiconductor. SnO<sub>2</sub> and In<sub>2</sub>O<sub>3</sub>, which have been attracting attention for their high detection performance, were used as sensing materials. The sensing material was easily deposited on the SiO<sub>2</sub> substrate in the form of nanorods through the e-beam evaporator GAD method. To improve detection performance and impart selectivity, Pt, Au, and Pd were deposited on the nanorod

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as catalysts. The fabricated sample was measured for  $\text{CH}_3\text{COCH}_3$ ,  $\text{C}_7\text{H}_8$ , and  $\text{NO}_2$ . High detection characteristics were confirmed at the sub ppb level.

## 2. EXPERIMENTAL

### 2.1 Fabrication process

Interdigitated electrodes (IDEs) were fabricated using Pt/Ti (150/30 nm), on a 4-inch  $\text{SiO}_2/\text{Si}$  substrate through photolithography. The distances between each electrode were 5  $\mu\text{m}$  and 20 electrodes were placed in an area measuring 1 mm  $\times$  1 mm. The Pt-IDEs patterned substrates were cleaned with acetone, ethanol, and deionized water, after which they were dried under a flow of nitrogen gas.

Subsequently, electron beam evaporation was performed to deposit the  $\text{SnO}_2$  and  $\text{In}_2\text{O}_3$  nanorods at a glancing angle of  $80^\circ$  at a rotation speed of 15 rpm in off-axis mode. Shadow masks were used to deposit exclusively onto the area containing the electrodes, and the substrate was placed 30 cm from the crucible. The base pressure and growth rates were  $5 \times 10^{-6}$  mTorr and  $1 \text{ \AA s}^{-1}$ , respectively. During the nuclei growth, the self-shadowing effect occurred depending on the incident angle of vapor flux. Next, the Pt, Au, and Pd catalysts were deposited by performing evaporation at a glancing angle of  $15^\circ$  with a rotation speed of 15 rpm in off-axis mode. The base pressure and growth rates were  $5 \times 10^{-6}$  mTorr and  $0.1 \text{ \AA s}^{-1}$ , respectively. All the fabricated specimens were annealed at  $550^\circ\text{C}$  for 2 h in ambient air after the process of deposition.

### 2.2 Characterization and gas sensor measurements

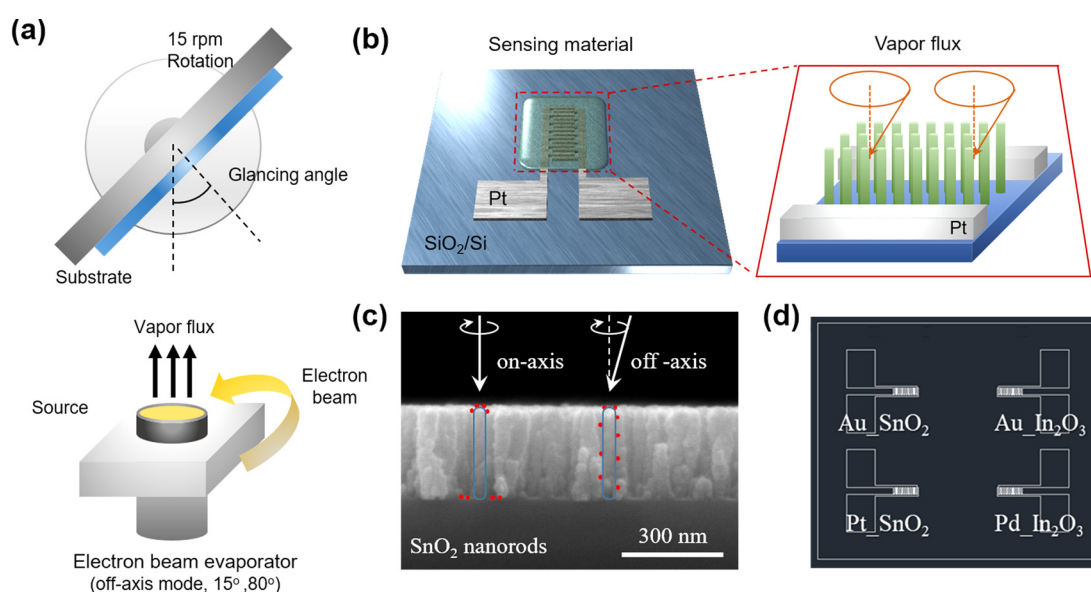
The morphology of the fabricated  $\text{SnO}_2$  and  $\text{In}_2\text{O}_3$  nanorods was observed using a field-emission scanning electron microscope (SEM) (Inspect F50) with an acceleration voltage of 15 kV and a working distance of 10 mm.

The gas sensing properties were measured by external heating of the quartz tube in a box furnace. The gas flow rate was maintained at 1000 sccm using mass flow controllers, and the gas flow was varied from dry air to a calibrated target gas. The resistances were measured at a DC bias voltage of 1 V using a source measurement unit (Keithley 2401), and all the measurements were recorded on a computer using LabVIEW with general purpose interface bus (GPIB).

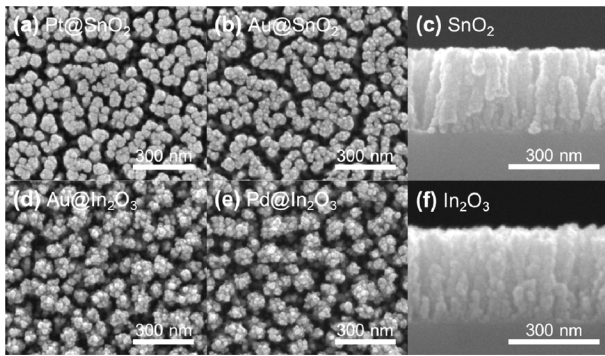
## 3. RESULTS AND DISCUSSIONS

### 3.1 Fabrication of nanorods and catalysts

The sensing materials,  $\text{SnO}_2$  and  $\text{In}_2\text{O}_3$ , are deposited directly on the PT IDE using the e-beam evaporator GAD method (Fig. 1(a)). At this time, it is possible to form well-aligned nanorods without a separate process due to the shadowing effect generated by the high tilt angle. The shadowing effect must be taken into consideration when evaporation equipment, such as e-beam evaporator with excellent straightness, is used, and the deposition process occurs at a high angle. It is deposited in the form of inclined nanorods, which can form straight nanorods through



**Fig. 1.** Schematic of (a) the fabrication procedures using GAD method via electron beam evaporation, (b) Nanorods deposited on Pt interdigitated electrode (IDE), and (c) catalyst decorated on nanorods using off-axis mode, (d) overall array gas sensor.



**Fig. 2.** Top-view SEM images of (a) Pt or (b) Au-decorated SnO<sub>2</sub> nanorod and (d) Au or (e) Pd-decorated In<sub>2</sub>O<sub>3</sub>. Cross-sectional SEM images of (c) SnO<sub>2</sub> nanorod and (f) In<sub>2</sub>O<sub>3</sub> nanorod.

substrate rotation. Next, the low-angle GAD method can also be used for Pt, Au, and Pd catalyst deposition (Fig. 1(d)). While using the existing on-axis method, the catalyst cannot be dispersed on the entire surface of the straight nanorods and can be concentrated only on the upper part of the nanorods. This problem can be solved by dispersing the catalyst from a low angle off-axis to the inside of the nanorods.

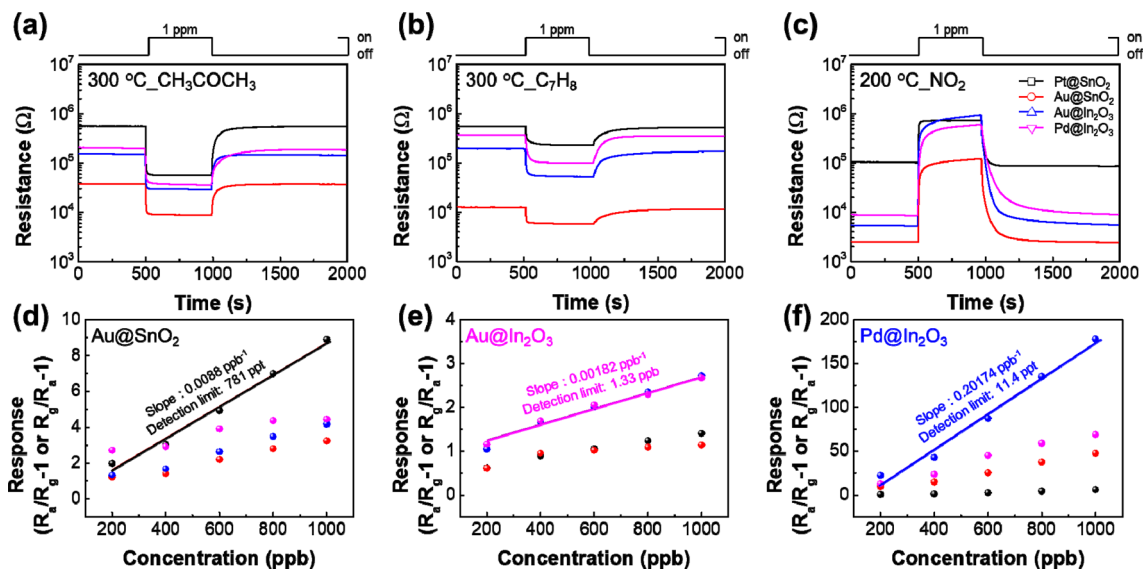
Figure 2 illustrates the top and cross-sectional SEM images of the well-aligned SnO<sub>2</sub> and In<sub>2</sub>O<sub>3</sub> nanorods according to the fabrication procedure. All samples deposited with the same length of 300 nm and the surface changes depending on the catalyst can be confirmed through the top image.

### 3.2 Gas sensing characteristics

The gas-sensing properties of individual samples were investigated by exposing all samples to 1 ppm of CH<sub>3</sub>COCH<sub>3</sub> at 300°C, C<sub>7</sub>H<sub>8</sub> or NO<sub>2</sub> at 200°C (Fig. 3(a)-(c)). The response is defined as  $(R_g/R_a - 1)$ , where R<sub>a</sub> and R<sub>g</sub> denote the resistance of the sensors in the absence and presence of the target gas, respectively. Each sample exhibited different sensing properties depending on the decorated catalyst. In particular, Au-decorated SnO<sub>2</sub> produced a response of 8.9 for 1 ppm CH<sub>3</sub>COCH<sub>3</sub>, Au-decorated In<sub>2</sub>O<sub>3</sub> displayed a response of 2.7 for 1 ppm C<sub>7</sub>H<sub>8</sub>, and a Pd-decorated In<sub>2</sub>O<sub>3</sub> displayed a sensitivity of 178 for 1 ppm NO<sub>2</sub>.

In addition, the gas detection characteristics were measured for various concentrations from 0.2–1 ppm (Fig. 3(d)-(f)) to measure the linearity and long-term stability of the sensor for each gas concentration. Through this, we confirmed through calculation that Au-decorated SnO<sub>2</sub>, Au-decorated In<sub>2</sub>O<sub>3</sub>, and Pd-decorated In<sub>2</sub>O<sub>3</sub> have detection limits of 781 ppt, 1.33 ppb, and 11.4 ppb respectively for CH<sub>3</sub>COCH<sub>3</sub>, C<sub>7</sub>H<sub>8</sub>, and NO<sub>2</sub>.

Humidity and measurement temperature are the factors that significantly influence the sensing characteristics of a sensor. To investigate the gas detection characteristics based on the moisture and temperature, we measured 5 ppm CH<sub>3</sub>COCH<sub>3</sub>, C<sub>7</sub>H<sub>8</sub>, and NO<sub>2</sub>, at various temperatures ranging from 150-350 °C at 40% relative humidity (Fig. 4). It was confirmed that the sensor sensitivity had high selectivity for a specific temperature and gas, and the sensitivity was slightly reduced compared to that in a dry environment, but it still had high selectivity and distinguishable



**Fig. 3.** (a–c) Real time response to 1 ppm CH<sub>3</sub>COCH<sub>3</sub>, C<sub>7</sub>H<sub>8</sub> and NO<sub>2</sub>. (d–f) Sensor response as a function of gas concentration and theoretical detection of limits at 300°C or 200°C in dry atmosphere.

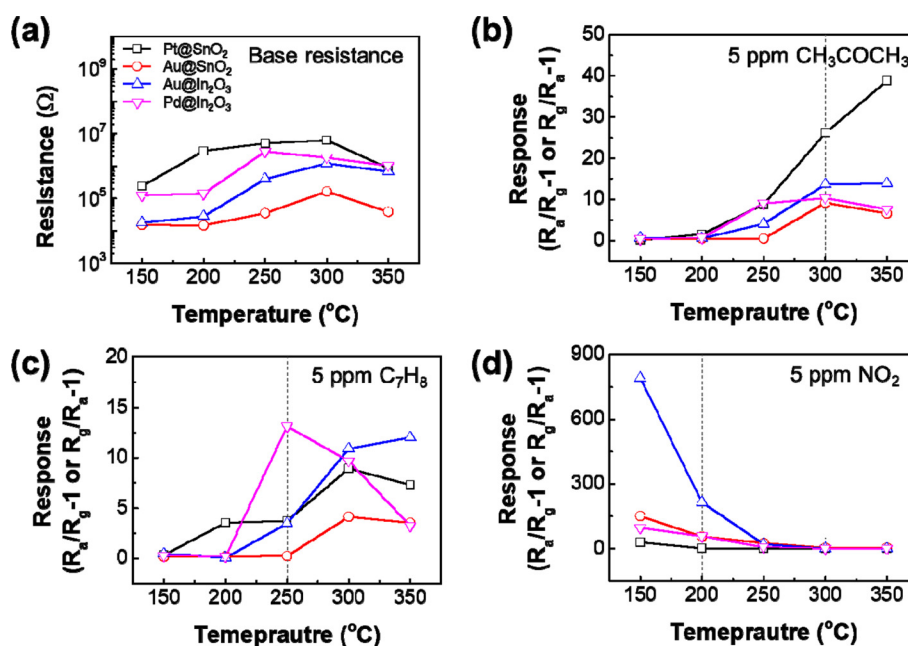
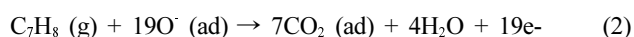
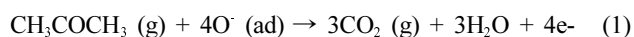


Fig. 4. Base resistance transients and response plot of SnO<sub>2</sub> nanorod and In<sub>2</sub>O<sub>3</sub> nanorod to 5 ppm CH<sub>3</sub>COCH<sub>3</sub>, C<sub>7</sub>H<sub>8</sub> and NO<sub>2</sub> as a function of operating temperature from 150-350°C in 40% relative humidity condition.

sensitivity.

### 3.3 Gas sensing mechanism

Oxygen adsorbed on the surface of the oxide semiconductor from the surrounding air forms a depletion layer and increases the resistance (Fig. 5(a)). When a catalyst is used, the catalyst is distributed over the entire surface of the semiconductor to move the electrons from the semiconductor to the catalyst, thereby promoting the formation of a depletion layer of the semiconductor and further increasing the resistance (Fig. 5(b)). At this time, oxygen species are adsorbed on the metal oxide surface in the form of O<sub>2</sub><sup>-</sup>, O<sup>-</sup>, or O<sup>2-</sup> depending on the temperature. Afterwards, the reducing gases, CH<sub>3</sub>COCH<sub>3</sub> and C<sub>7</sub>H<sub>8</sub>, react with oxygen on the surface of the semiconductor and cause the following changes:



This leads to desorbing oxygen on the surface of the semiconductor and reducing the resistance (Fig. 5(c)). In contrast, in the process wherein NO<sub>2</sub> is adsorbed on the semiconductor surface in the same way as oxygen, electrons are extracted from the semiconductor surface, forming a depletion layer and

increasing the resistance (Fig. 5(d)). In the case of NO<sub>2</sub>, the adsorbed oxygen has a higher reactivity at low temperatures because the adsorbed oxygen limits the space in which NO<sub>2</sub> is adsorbed.

## 4. CONCLUSIONS

To solve the existing low selectivity problem, we simultaneously observed the sensing characteristics of various sensors in the form of an array. We deposited SnO<sub>2</sub> and In<sub>2</sub>O<sub>3</sub> in the form of nanostructures directly on the IDE without a separate process through the e-beam evaporator GAD method. At this time, Pt, Au, and Pd were used as catalysts to enhance the sensing characteristics. The prepared samples displayed high detection characteristics of sub ppb level for CH<sub>3</sub>COCH<sub>3</sub>, C<sub>7</sub>H<sub>8</sub>, and NO<sub>2</sub> gases in a dry environment. It was confirmed that these high detection characteristics can be distinguished even in a humid condition. In addition, the selective reaction between gases, which is difficult to distinguish when using a single sample, was also supplemented by using multiple sensors. This study highlights the advantages of fabricating one-chip array sensors, and we are confident that using an array that integrates a greater number of sensors can provide high selectivity.

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